

Nanostructured Materials Produced by Severe Plastic Deformation

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- Severe plastic deformation of metals leads to an extremely fine microstructure. This is produced by “fragmentation” of the original crystal structure. The mechanisms for this fragmentation are discussed. Various methods for severe plastic deformation are classified according to the strain paths involved:
 - ✓ Continuous strain without change of strain path
 - ✓ Accumulated strain without change of strain path
 - ✓ Accumulated strain with reversal of strain path
 - ✓ Accumulated strain with variable strain path
- For comparison of different experiments, the concepts of “equivalent strain” and “strain efficiency” are discussed.

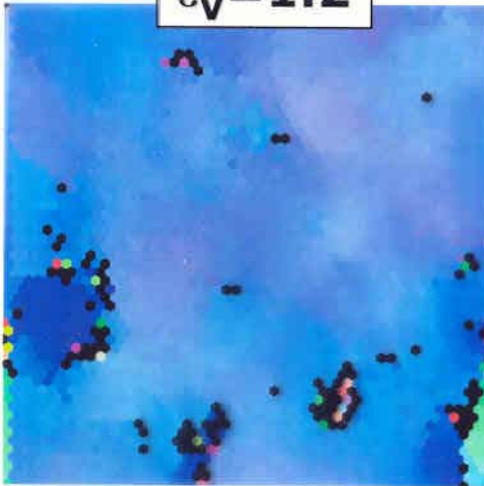
SPD

Continuous strain no change of strain path	Compression Extrusion Torsion (HPT)
Accumulated strain no change of strain path	Rolling Drawing ECA (route A)?
Accumulated strain reversal of strain path	CEC ECA (route C) CCDC
Accumulated strain variable strain path	Swaging ECA (other routes) accumulated roll bonding

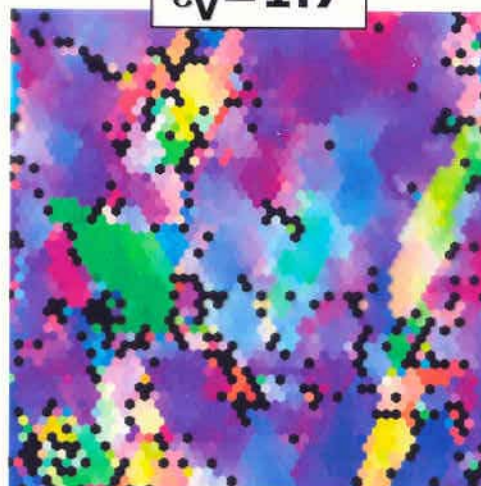
equivalent strain?

evolution of microstructure at large strains in Cu

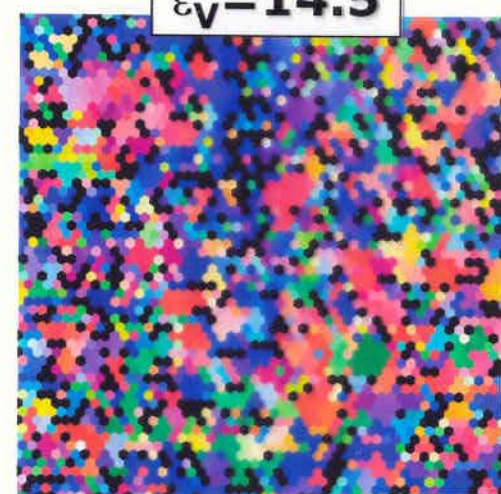
$\varepsilon_v = 1.2$



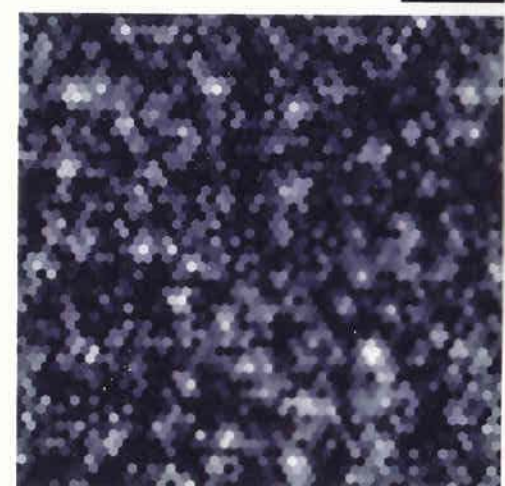
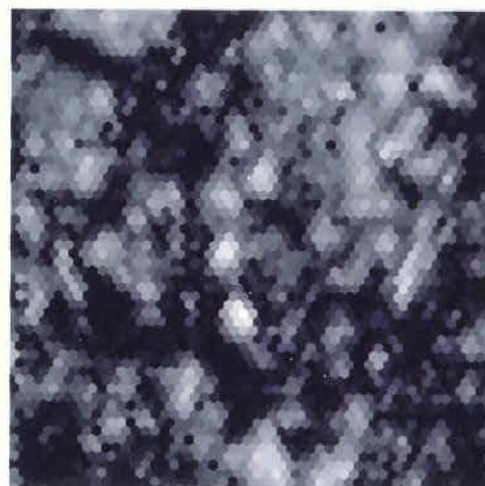
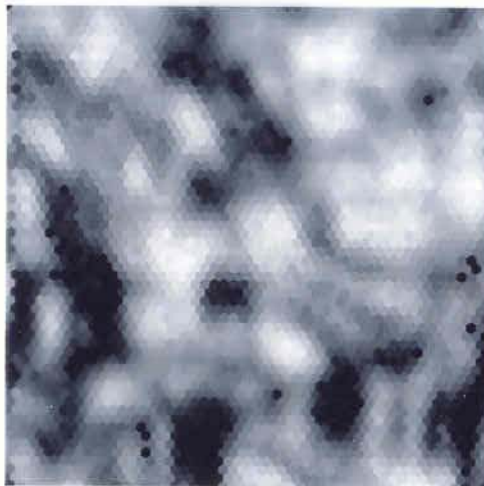
$\varepsilon_v = 1.7$

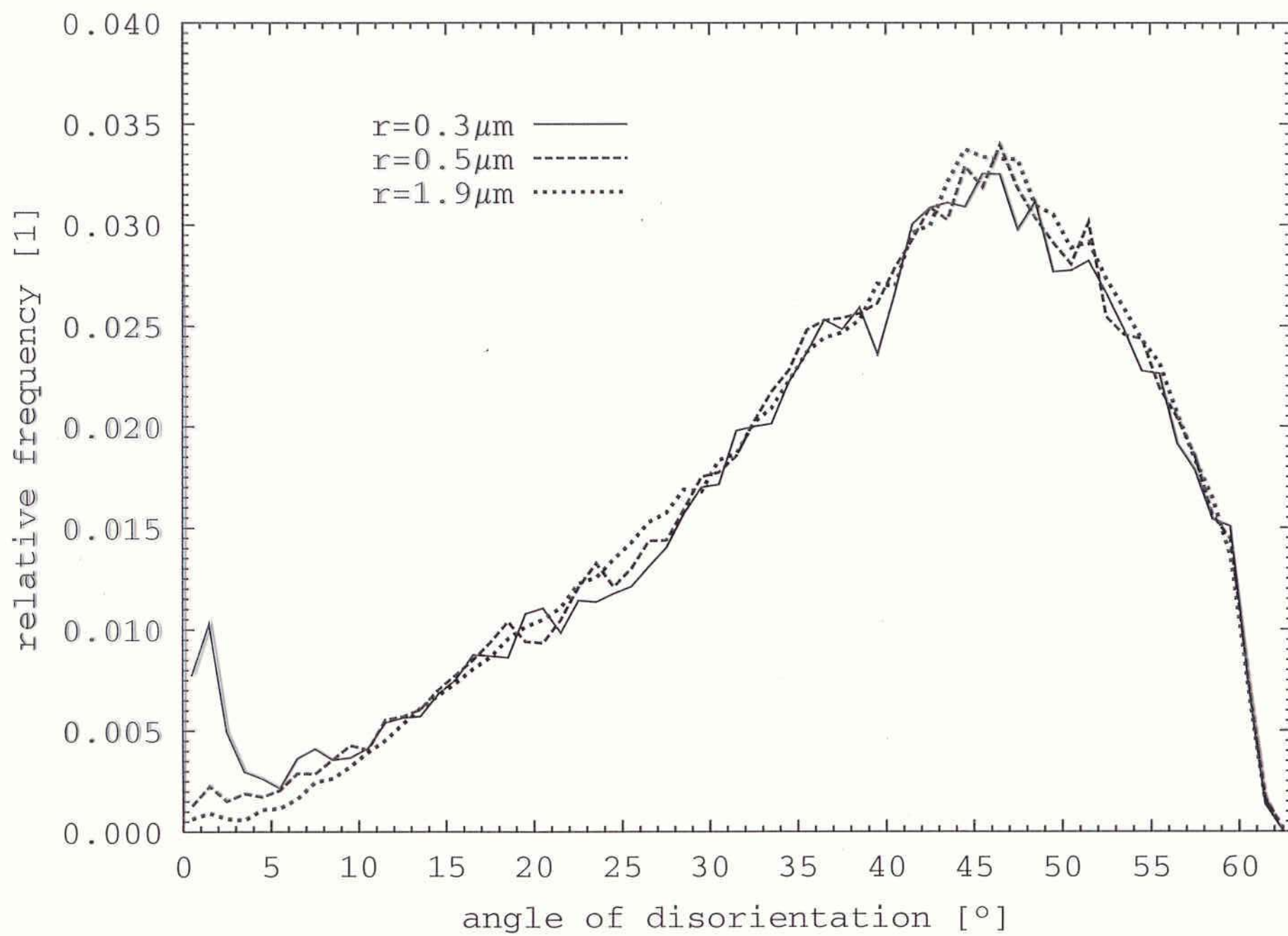


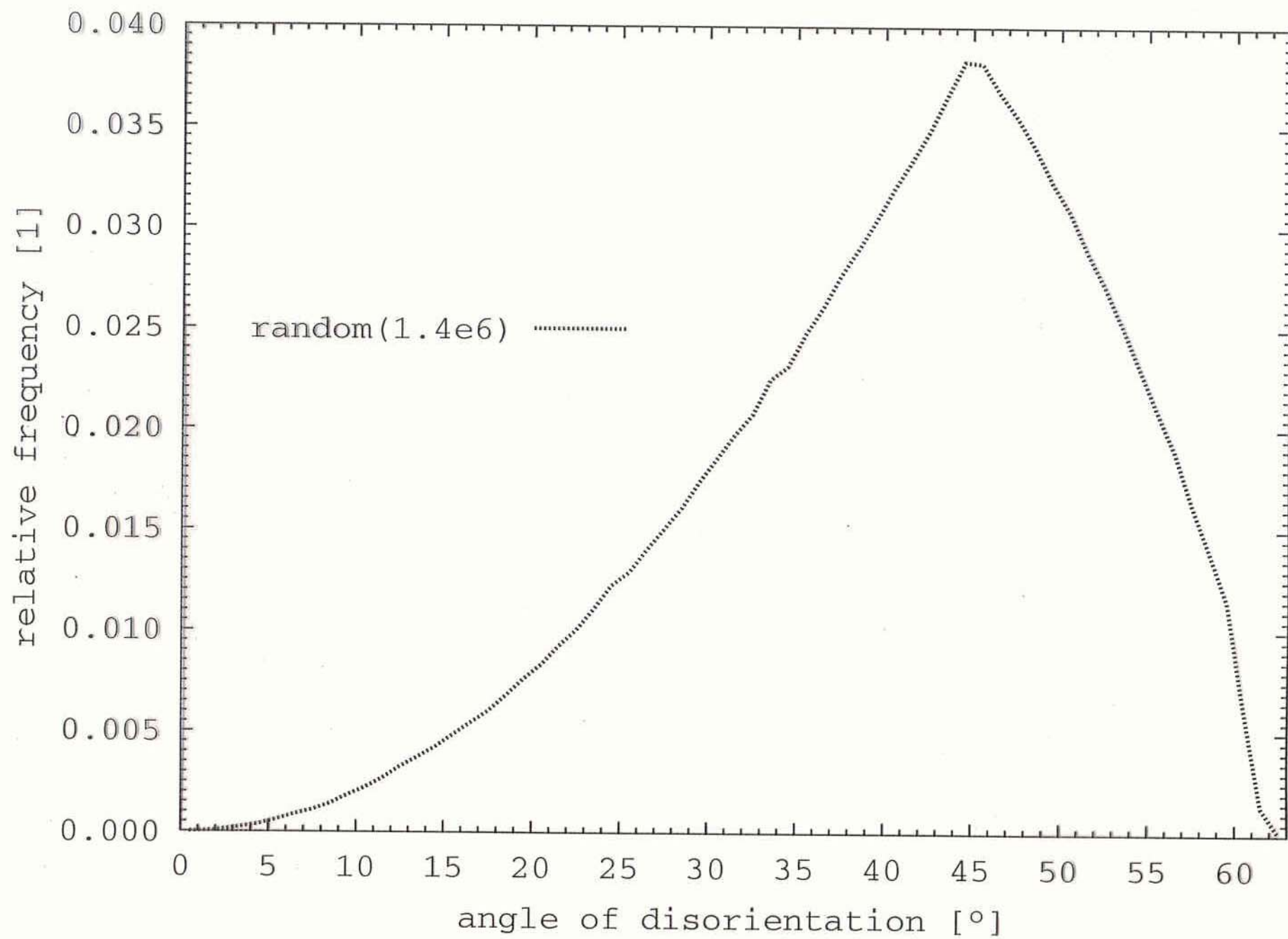
$\varepsilon_v = 14.5$

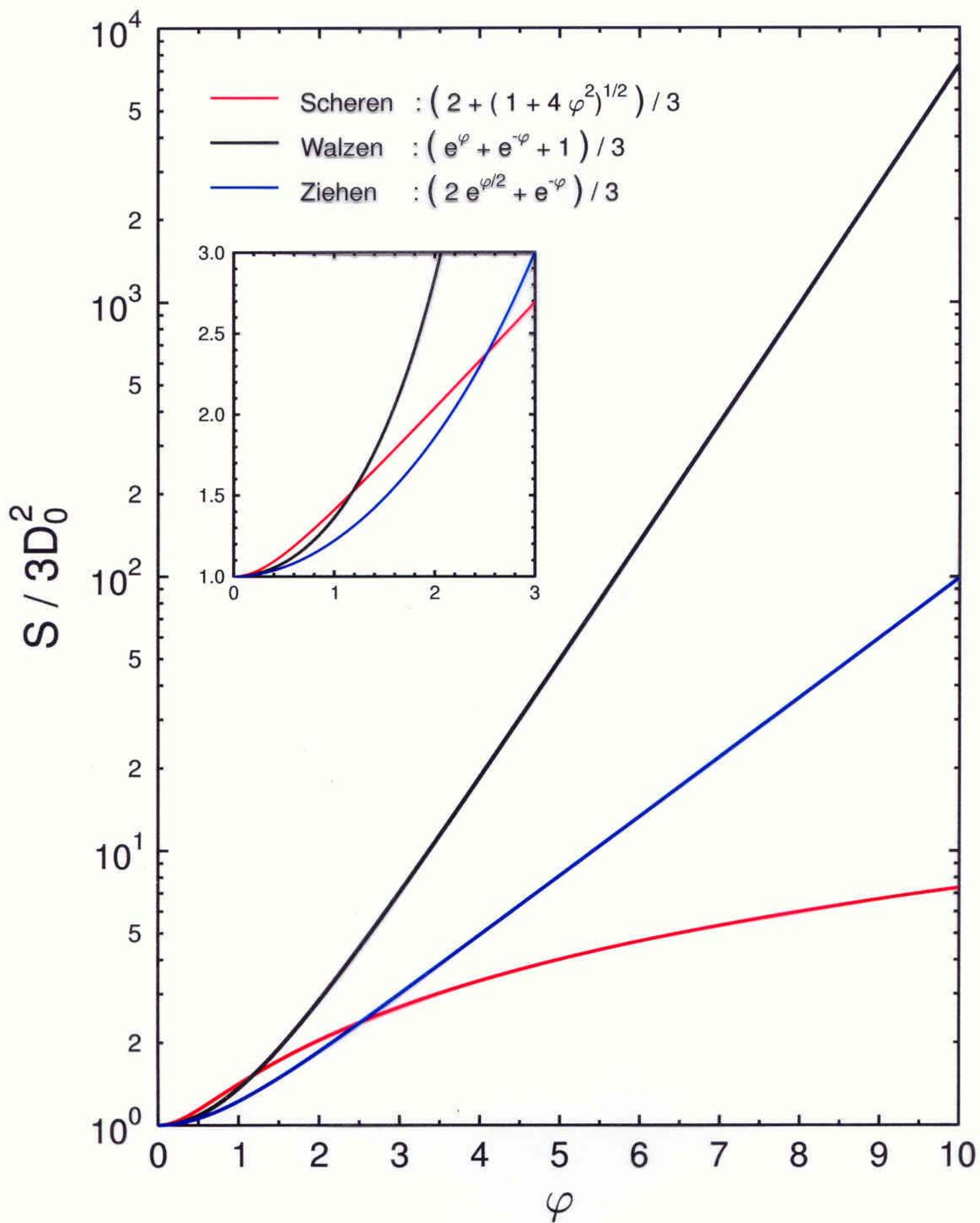


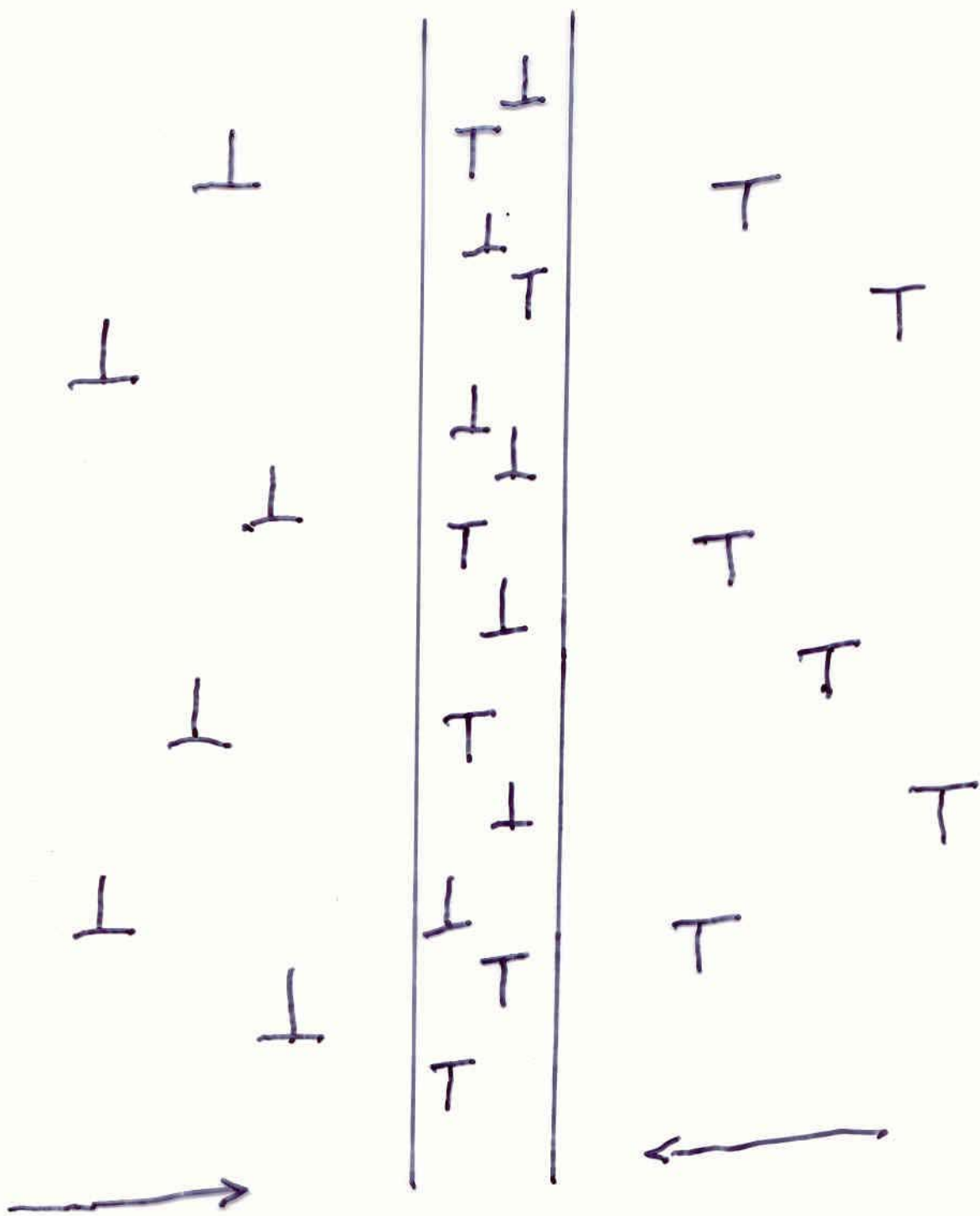
2 μm











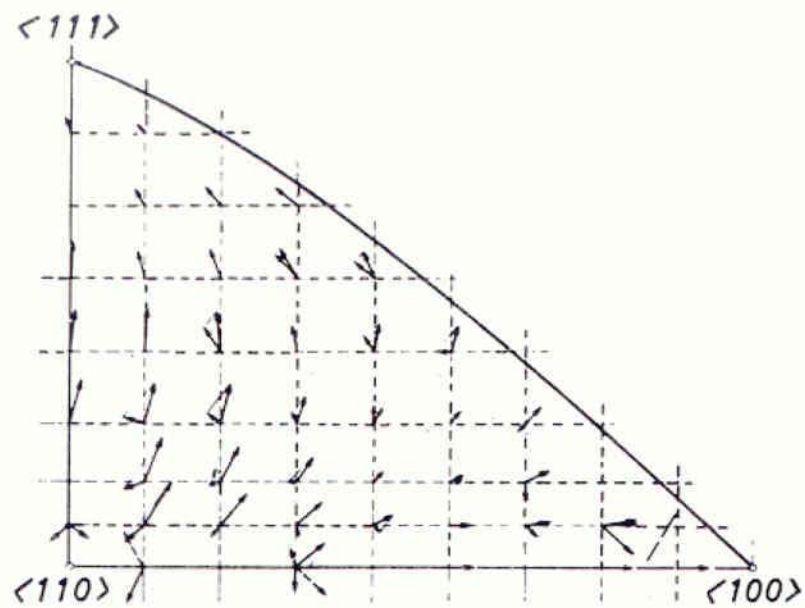
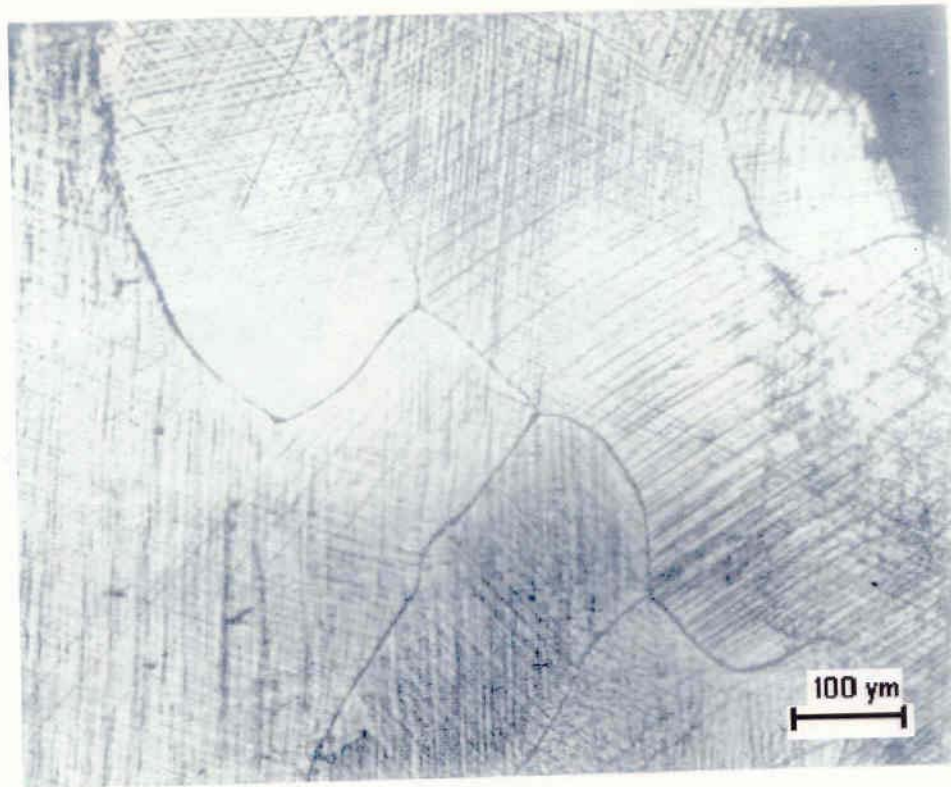
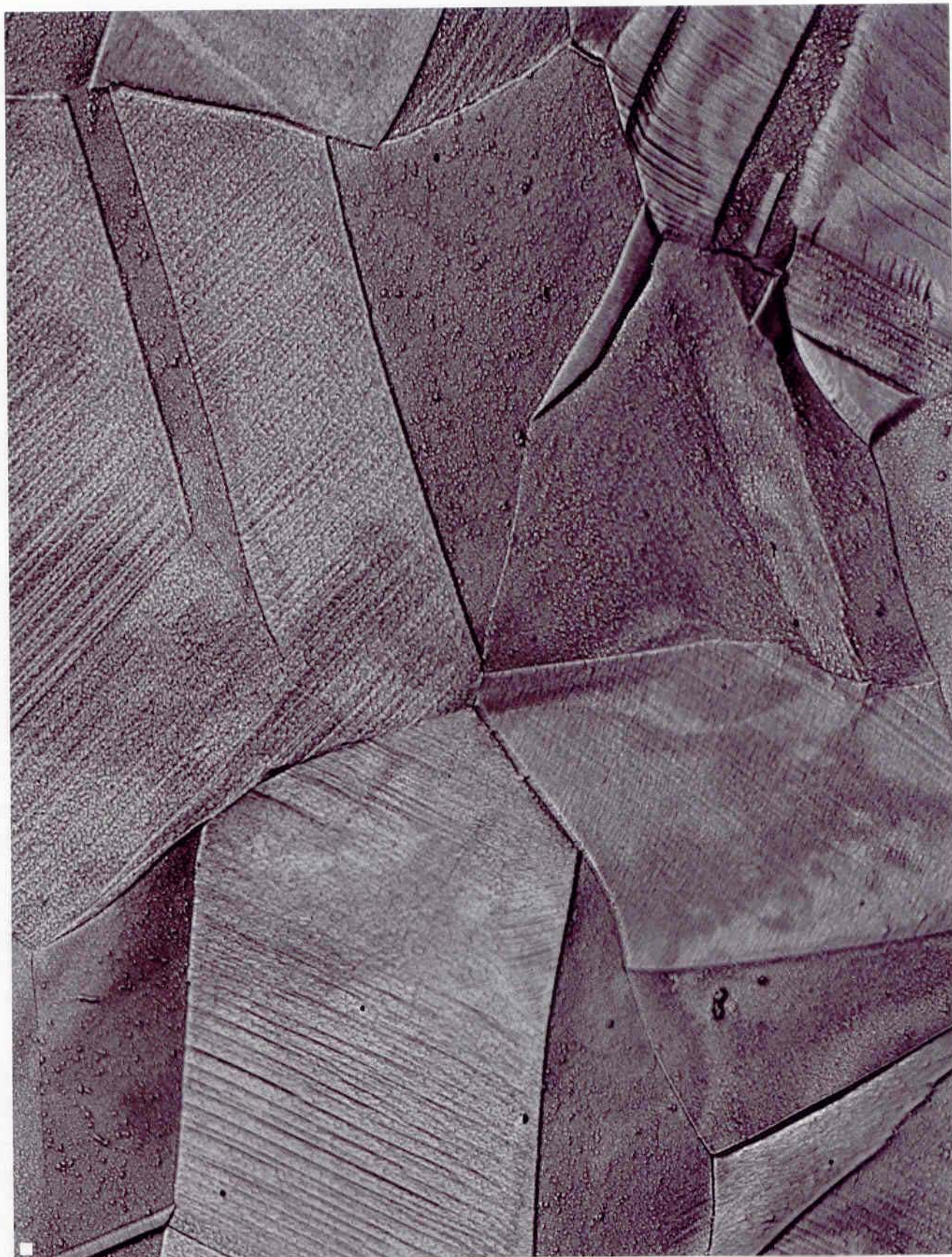


Fig.1. Lattice rotations for axisymmetric tension, determined by TAYLOR [8] for $\{111\}\langle 110 \rangle$ slip. The sense of rotation is reversed for compression;
2.37% strain



Blei



Tension

$$\frac{A}{V} = \int \sigma \frac{dl}{l} = \int \tilde{\sigma} dy$$

Simple Shear

$$\frac{A}{V} = \int \tau dy$$

Equivalent Strain for

$$\int \tilde{\sigma} dy = \int \tau dy$$

fulfilled for

$$\tilde{\sigma} = a \tau \quad \text{and} \quad dy = \frac{1}{a} d\gamma$$

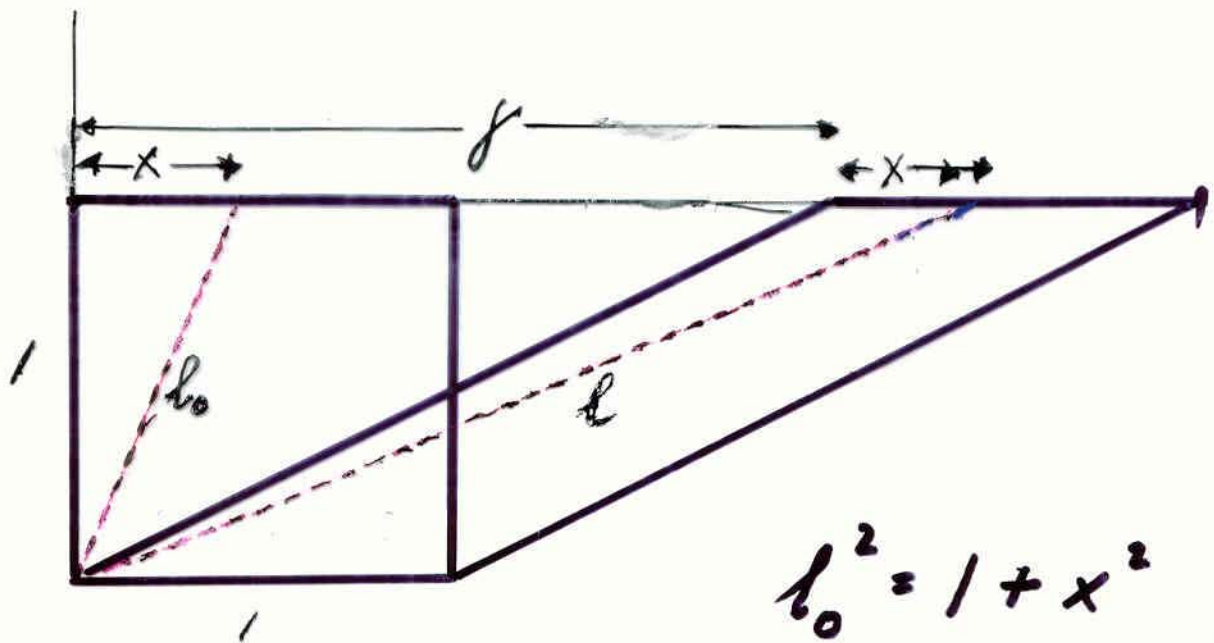
↓

Tresca: $a=2$

v. Mises: $a=\sqrt{3}$

↓

?



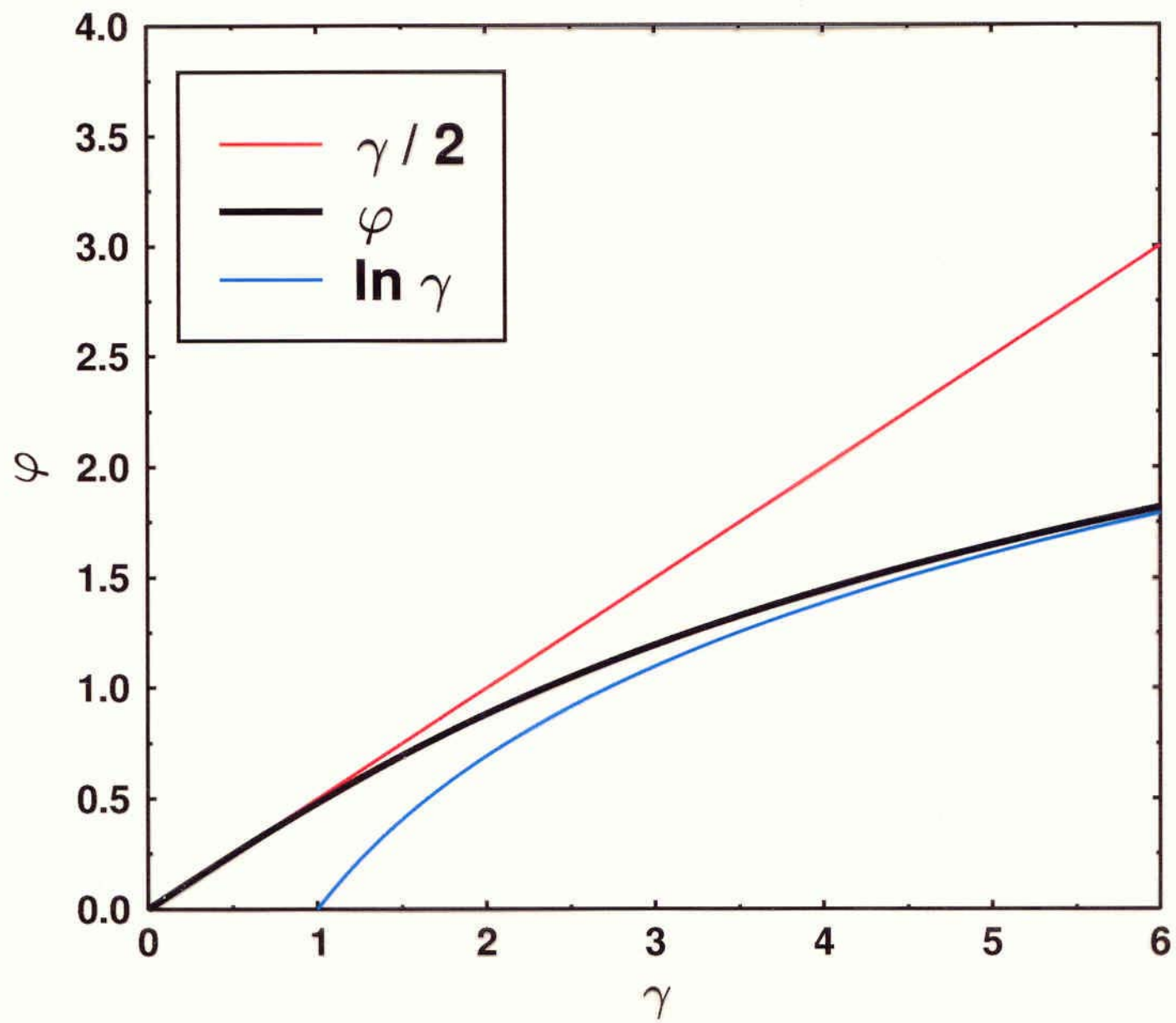
$$l_0^2 = 1 + x^2$$

$$l^2 = 1 + (f+x)^2$$

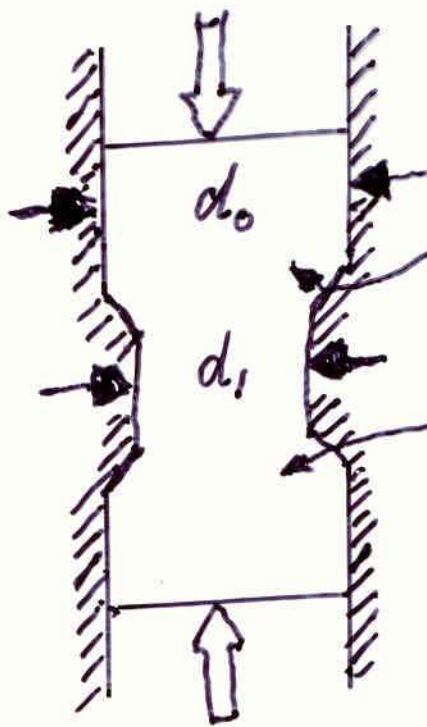
$$\frac{d}{dx} \left(\frac{l}{l_0} \right)^2 = 0 \rightarrow x = \sqrt{\frac{f^2}{4} + 1} - \frac{f}{2}$$

$$y = \ln \frac{l}{l_0}$$

$$\ln \frac{l}{l_0} = \frac{1}{2} \ln \frac{1 + \left(\sqrt{\frac{f^2}{4} + 1} + \frac{f}{2} \right)^2}{1 + \left(\sqrt{\frac{f^2}{4} + 1} - \frac{f}{2} \right)^2}$$



Reversal of strain path CEC



$$\Delta \varphi_1 = \ln \left(\frac{d_0}{d_1} \right)^2$$

$$\Delta \varphi_2 = -\Delta \varphi_1$$

$$\varphi_{equ} = \sum_1^{2n} |\Delta \varphi_n| \quad ?$$

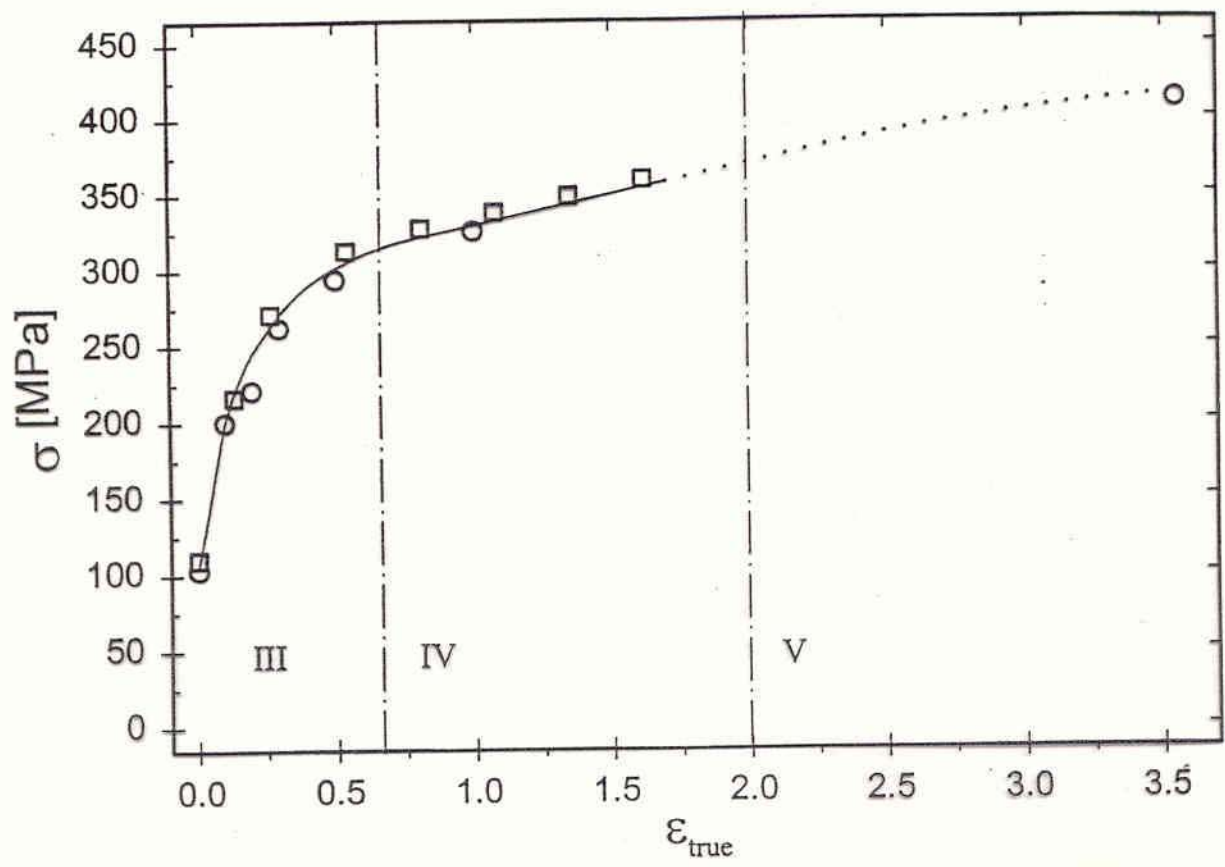
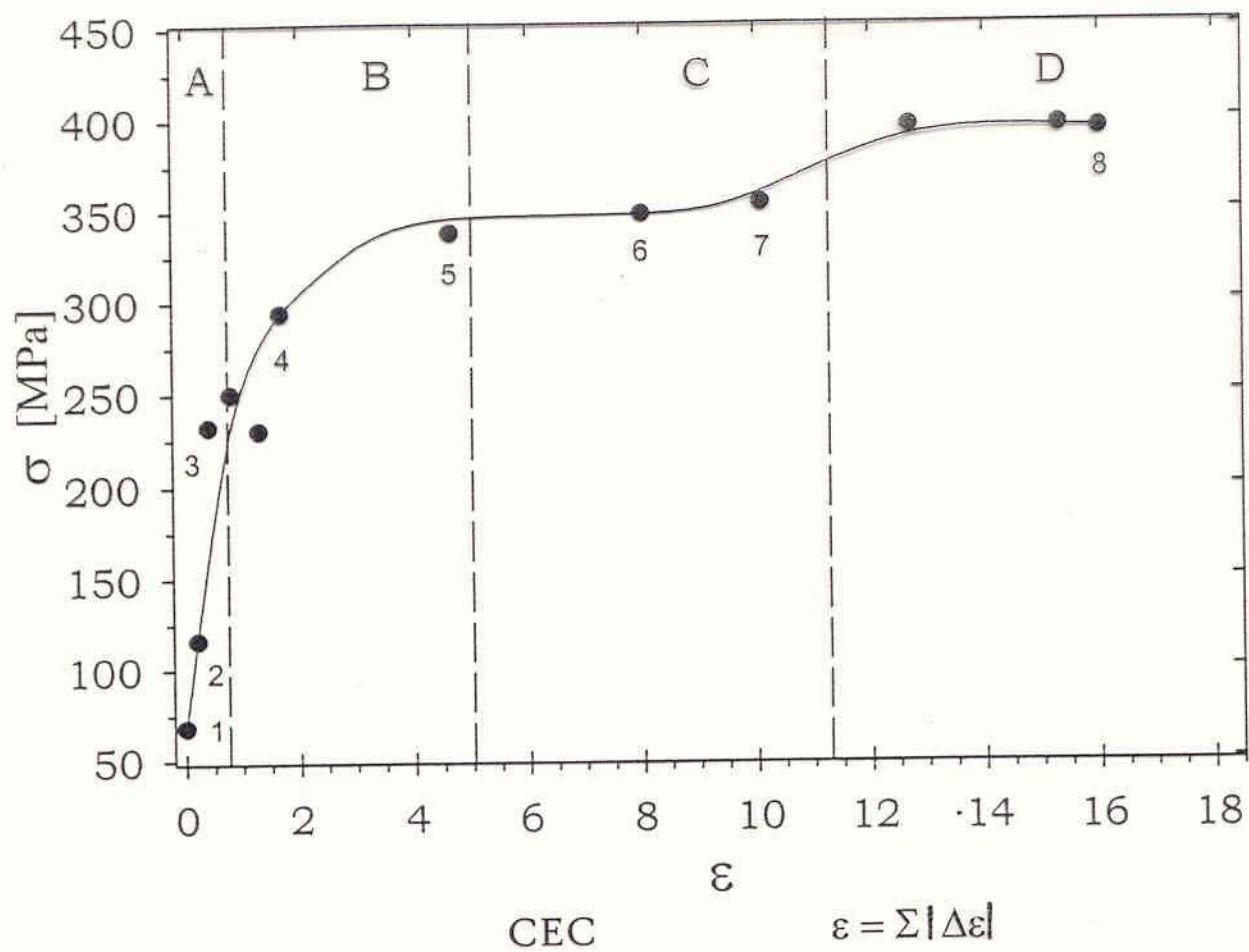
$$\varphi_{equ} = \eta \sum_1^{2n} |\Delta \varphi_n|$$

$$\frac{d\varphi}{d\gamma} \approx \frac{2M}{\Lambda b} \quad \begin{array}{l} \text{recovery} \\ \text{dynamic} \end{array} \quad \begin{array}{l} \text{"geometric"} \end{array}$$

$$\gamma = 2\varphi \rightarrow a = \frac{d\gamma}{d\varphi} = 2$$

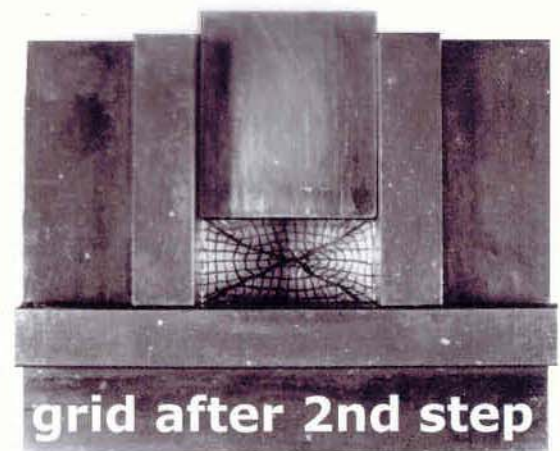
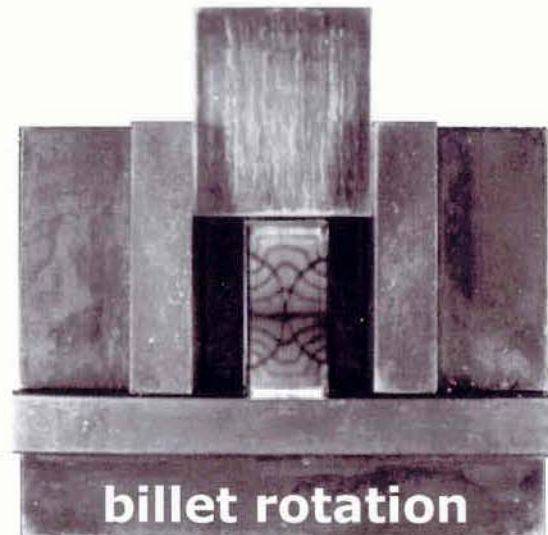
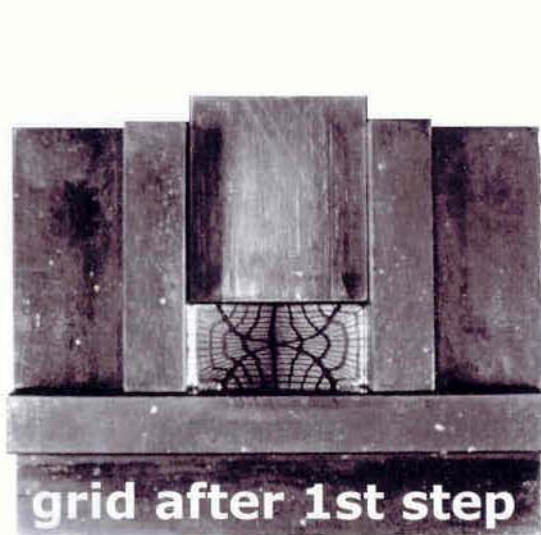
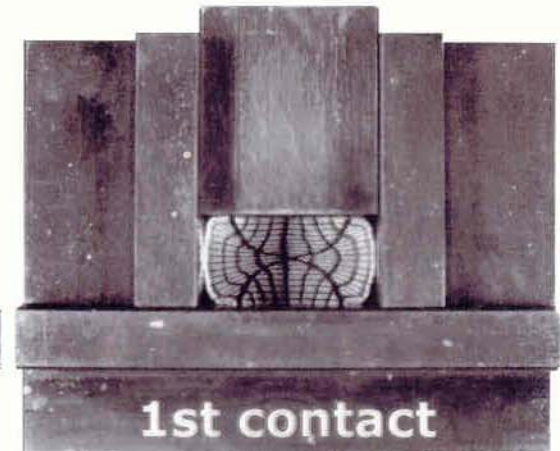
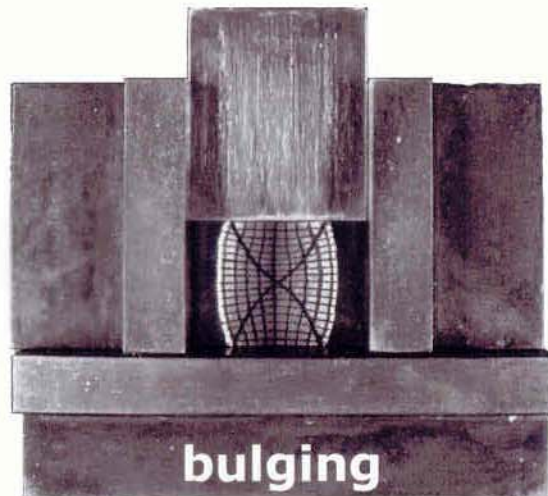
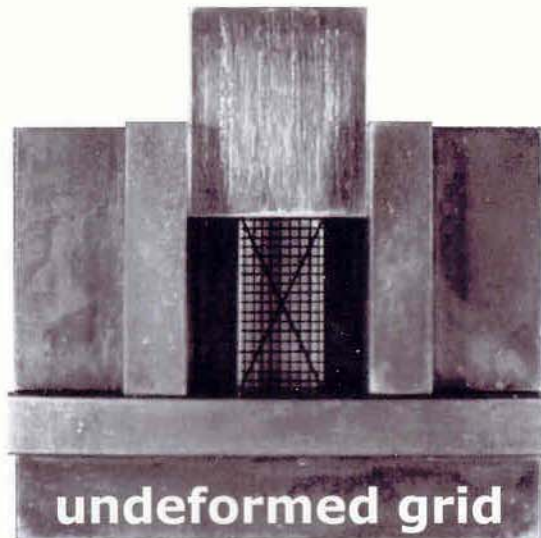
$$\gamma = e^\varphi \rightarrow a = \frac{d\gamma}{d\varphi} = \gamma$$

AlMg5

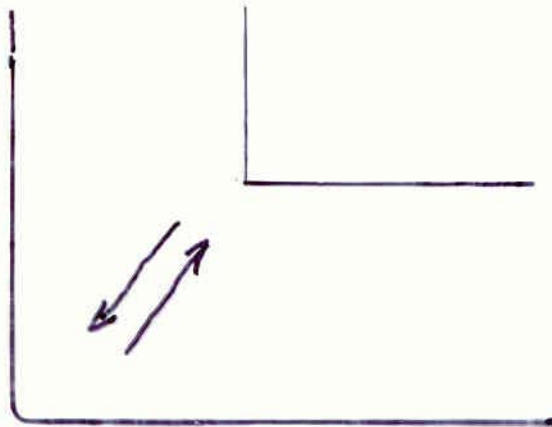
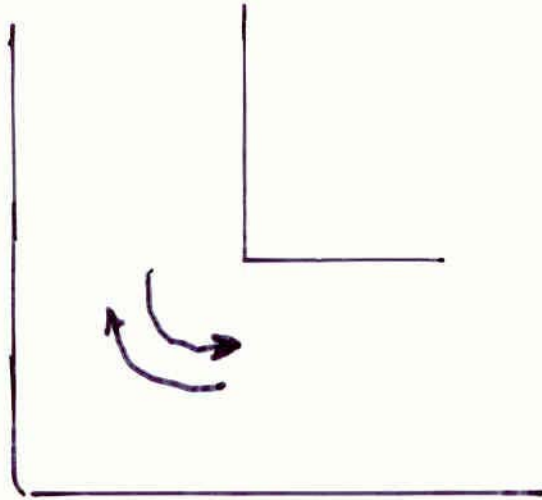


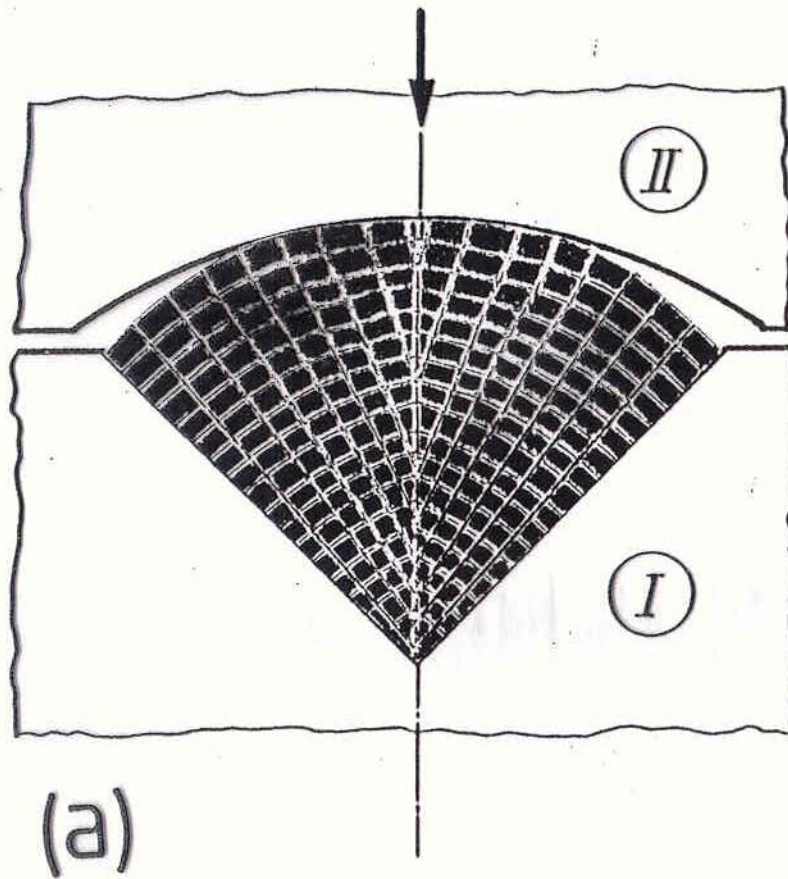
unidirectional strain

Cyclic Channel Die Compression (CCDC)

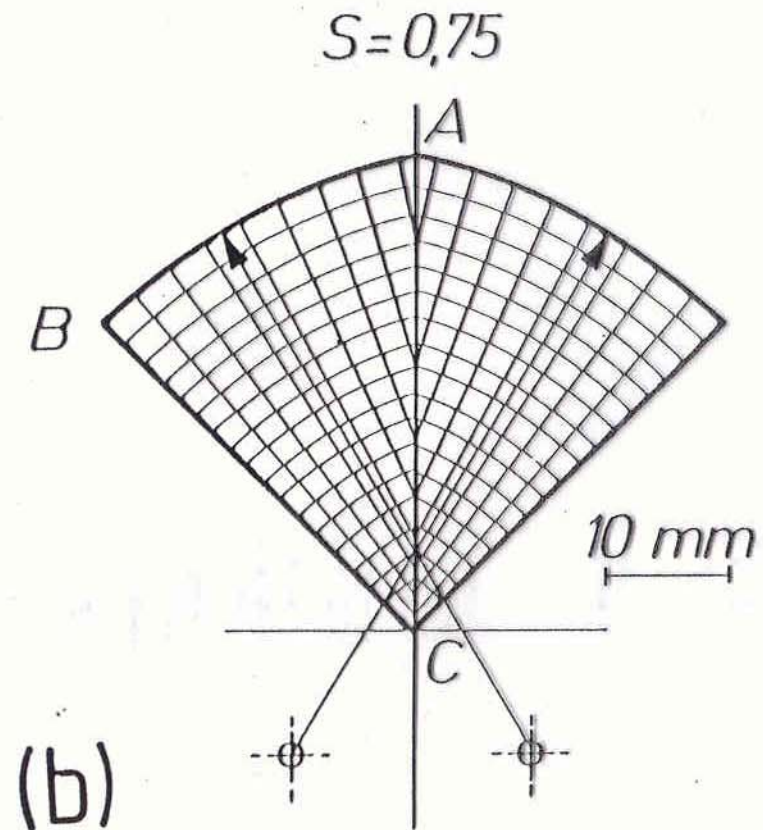


ECA



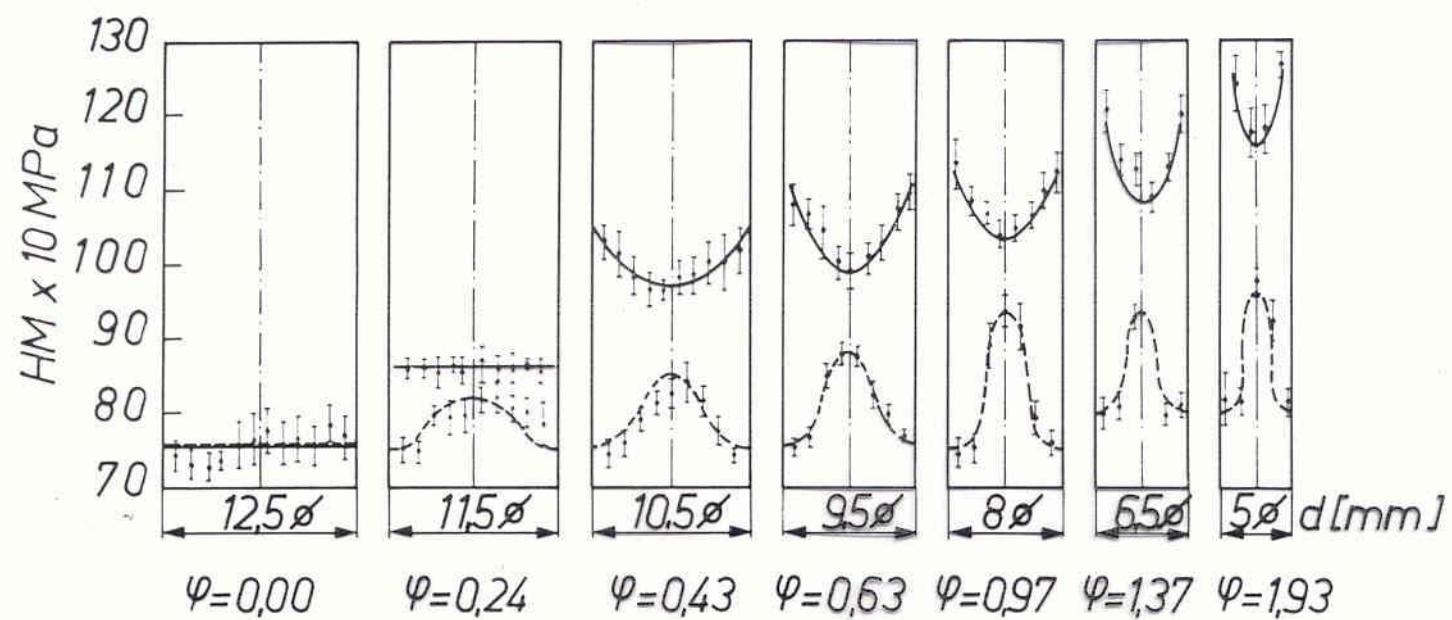


a distorted grid after partial deformation;
scratched-in polar coordinate grid



b lead testpiece with

4 Visioplastic experiment



CONCLUSIONS

- SPD produces microstructures with very small structural elements separated by interfaces built up of many dislocations produced by straining. The process of fragmentation is discussed.
- For a comparison of the efficiency of different SPD techniques to produce such structures the traditional concept of “equivalent strain” based on equal mechanical work is problematic, especially when strain is accumulated in steps with varying strain path where “geometric recovery” must be taken into account.